Influence of Hydrogen Dilution of Silane on the Properties of nc-Si:H films grown by Layer-by-Layer deposition technique

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Abstract. Hydrogenated nanocrystalline silicon (nc-Si:H) films produced by layer-by-layer (LBL) deposition technique were studied. The films were grown at different hydrogen to silane flow-rate ratio on crystal silicon (111) substrate. The properties of films were investigated by X-ray diffraction (XRD), micro-Raman scattering spectroscopy, Fourier transform infrared (FTIR) spectroscopy, optical transmission spectroscopy, atomic force microscopy (AFM) and field emission scanning electron microscopy (FESEM). These properties showed dependence on the hydrogen dilution of silane. Appearance of XRD peaks at diffraction angles of 28.4 ° and 56.1 ° which correspond to silicon orientation of (111) and (311) respectively, were observed in all films indicating evidence of crystallinity in the films. Raman scattering results indicated that crystallinity in the films was due to the presence of nanocrystallites embedded in an amorphous matrix. The energy gap of the films showed dependence on the hydrogen content in the films. Increase in nanocrystallite size resulted in increase in disorder at low hydrogen dilution films but films remain homogenous with increase in nanocrystallite size for the high hydrogen dilution films.

Introduction

Hydrogenated nanocrystalline silicon (nc-Si:H) thin films, with nanometer sized crystallites embedded within an amorphous matrix demonstrate superior properties for applications such as solar cells and thin film transistors (TFTs) [1]. Solar cells fabricated using nc-Si:H are more stable during light soaking while TFTs are more efficient due to enhancement of carrier mobility. The role of hydrogen is important to the properties of nc-Si:H and proposed growth model of nc-Si:H such as surface diffusion and hydrogen etching models lead to a clearer understanding of the role of hydrogen in nc-Si:H [2]. In this work, we report our investigations on the effects of hydrogen dilution of silane on the properties of nc-Si:H films prepared by layer-by-layer (LBL) deposition technique using X-ray diffraction (XRD), micro-Raman scattering spectroscopy, Fourier transform infrared (FTIR) spectroscopy, optical transmission spectroscopy, atomic force microscopy (AFM) and field emission scanning electron microscopy (FESEM).

Methodology

The nc-Si:H films were deposited on silicon (111) substrates using LBL technique in a conventional home-built radio-frequency plasma enhanced chemical vapor deposition (rf PECVD) system. During the LBL deposition, the silane (SiH₄) flow-rate was fixed at 5 sccm while the hydrogen (H₂) flow-rate was varied for each deposition process to obtained films with different H₂ to SiH₄ flow-rate ratios. The LBL technique involved alternating the growth of nc-Si:H films from the discharge of H₂ mixed with SiH₄ for 5 minutes followed by H₂ plasma treatment on the pre-grown layer for 3 minutes for a period of 64 minutes (8 cycles). The substrate temperature and pressure were maintained at 300 °C and 0.8 mbar respectively. The 13.56 MHz rf power density was fixed at 1400 mW/cm² throughout the deposition process for all films. The optical transmission spectra of the films were measured using a UV-VIS-NIR spectrophotometer and from these data the optical

energy gap, E_G was estimated [3]. XRD and micro-Raman spectroscopy using 514.5 nm Argon laser line were employed to investigate the crystallinity of these films. The average crystallite size, d_{XRD} of the crystallites in the films was estimated using Scherrer's formula [4]. The crystalline volume fraction, X_C was determined from the integrated intensities of Raman peaks at 520, 500 and 480 cm^{-1} which corresponds to crystalline components, grain boundaries and amorphous components respectively [5]. The hydrogen content, C_H and the ratio of integrated intensities of the Si-H₂ to Si-H absorption bands were determined from FTIR spectra [6]. Surface morphology of the film was studied using AFM and FESEM.

Fig. 1. XRD spectra of nc-Si:H films at different hydrogen to silane flow-rate ratio, R.

Fig. 2. Average crystallite size, d_{XRD} and crystalline volume fraction, X_C of nc-Si:H film versus hydrogen to silane flow-rate ratio, R.

Results and Discussion

Appearance of sharp diffraction peaks corresponding to Si(111) and Si(311) at 28° and 56° respectively (Fig. 1) shows the presence of crystalline phase in the structure of all films deposited at different hydrogen to silane flow-rate ratio, R. Fig. 2 illustrates the variation of average crystallite size, d_{XRD} and crystalline volume fraction, X_C with R. It is observed that d_{XRD} increases to a maximum (60 nm) with increase in R from 2 to 4, then it decreases to a minimum (10 nm) when R increases to 8 followed by increase in d_{XRD} with further increase in R. The trend is the same for the variation of X_C with R except that X_C decreases to a minimum at R equals to 12. High X_C films contain higher crystalline component by fraction compared to the amorphous component. The films formed with R equals to 2 and 8 have small nanocrystallites embedded sparsely within the amorphous matrix of the film. Film with high X_C of 0.9 and with large nanocrystallites of 60 nm is formed with R equals to 4. Film with R equals to 12 has low X_C with 30 nm sized nanocrystallites embedded within a highly amorphous structure. The film prepared with R equals to 16 is truly a two-phase material with reasonably sized nanocrystallites of 40 nm with X_C of 0.6. The plots of Hydrogen content, C_H and optical energy gap, E_G versus R are shown in Fig. 3. C_H decreases to a minimum when R equals to 4 and then increases up to a maximum saturation value of 10 % when R is equal to 12. E_G shows similar dependence on R as C_H but E_G continues to increase with increase in R beyond R equal to 12. The plots indicate that E_G has some dependence on C_H . Increase in R reduces C_H and E_G for films with low R (R = 2, 4), while high R (R = 8, 12 & 16) films produces a reversed dependence. Fig. 4 shows the variation of the ratio of the integrated intensities of SiH₂ to SiH absorption bands, I_{SiH_2}/I_{SiH} with R. I_{SiH_2} and I_{SiH} is proportional to the concentration of SiH₂ and SiH bonds in the film respectively. The plot of d_{XRD} with R is inserted in Fig. 4 to show the relationship between these parameters. High I_{SiH_2}/I_{SiH} roughly indicates increase in structural disorder in the film structure. Films prepared with low R become more disordered with increase in nanocrystallite size while for high R films crystallite size has very little effect on the homogeneity of the film structure. Large R films and films with small nanocrystallites are more ordered

comparatively. The AFM and SEM images of films prepared at R values of 4 and 16 confirm the results above as shown in Fig. 5. The film prepared with R equals to 4 has clusters of nanocrystallites of varied sizes (large and small) and are unevenly distributed. This explains the less ordered structure of the film. The images of the film with R equal to 16 shows a homogeneous film structure with evenly distributed nanocrystallites.

Fig. 3. C_H and E_G of nc-Si:H films versus hydrogen to silane flow-rate ratio, R.

Fig. 4. Variation of d_{XRD} and I_{2100}/I_{2000} ratio with hydrogen to silane flow-rate ratio, R.

Fig. 5. AFM and field emission SEM images of nc-Si:H films at hydrogen to silane flow-rate ratio, R of 4 (a) and 16 (b).

Conclusions

Hydrogen to silane flow-rate ratio has significant influence on the H content and crystallite size in the nc-Si:H film produced using the LBL technique which directly influences the energy gap, crystalline volume fraction and the homogeneity of the film. The H plasma treatment on the pre-grown surface of the nc-Si:H layer grown by LBL technique increases crystallite size and homogeneity in low and high hydrogen to silane flow-rate ratio films respectively.

References

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