

Generation and Growth of Atomic-scale Roughness at Surface and Interface of Silicon Dioxide Thermally Grown on Atomically Flat Si Surface

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Abstract. Atomic force microscopy measurements are carried out on the surface and interface of SiO₂ thermally grown on an atomically flat Si surface, and the uniformity of thickness on an atomic scale is investigated. Protuberances on the surface of SiO₂ grown at 800 °C in 3.8 % O₂ diluted by argon at the atmospheric pressure are generated in the initial stage of oxidation and subsequently increases in height at the same positions. The surface RMS value of SiO₂ linearly increases up to the thickness of approximately 5 nm, which almost corresponds to what was called the initial oxidation thickness. This fact indicates that the roughness increase is related to the re-oxidation of SiO emitted from the Si/SiO₂ interface during the initial oxidation. Additionally, the surface RMS value is saturated at a value of approximately 0.13 nm. A correspondence between the position of the dimples on the surface and the positions of the protuberances at the interface is also clear. The fact is the direct evidence of the local thinning of the thermally grown SiO₂ films. Based on these results, an atomic scale thermal oxidation mechanism is discussed.

Introduction

A non-uniform gate dielectric film can induce a fluctuation in the threshold voltages or lifetimes of time-dependent dielectric breakdown of ultra-fine metal-insulator-semiconductor field-effect-transistors (MISFET's). It is very important to clarify the underlying generation mechanism of such thickness variations. On the other hand, dielectric breakdown is also a local phenomenon: the dielectric film breaks down at its weakest point. To realize the higher reliability of the dielectric films, a more uniform film thickness on an atomic-scale is needed. For example, for the dielectric films with a thickness of approximately 3nm, a variation of a few atomic layers is equivalent to approximately 10 % of the total film thickness. Such film thickness variation also induces a fluctuation in the direct tunnel leakage current [1].

The Deal-Grove model is well known for thermal oxidation of silicon [2]. In this model, the oxidizing-species diffuse across a preexisting oxide layer, and react at the silicon surface, and a new

SiO₂ is formed there. A volume enhancement of 2.3 times at the interface induces mechanical stress that affects various aspects of thermal oxidation.

On the other hand, it has been reported that thermal oxidation of an atomically flat Si under lower oxygen pressure progresses in a layer-by-layer manner in a SiO₂ film thickness of a few atomic layers [3]. If the layer-by-layer oxidation is strictly maintained, a similar topography should be produced at the Si/SiO₂ interface during thermal oxidation. A precise evaluation of the oxidation process has shown that although thermal oxidation at the atmospheric pressure can be thought of as a layer-by-layer process, strictly speaking, it proceeds in a multilayer manner [4, 5]. Neither the roughening mechanism nor the thickness uniformity during thermal oxidation at the atmospheric pressure has so far been clarified.

In the present study, both issues were investigated by precise observations of the topography at the SiO₂ surface and Si/SiO₂ interface using atomic force microscopy (AFM).

Experimental

Czochralski-grown (111)-oriented mirror-polished Si wafers were used. After modified RCA cleaning, the wafers were immersed in ultralow dissolved-oxygen water (LOW). The typical root mean square (RMS) value at the surface of a commercial-base mirror-polished silicon wafer is more than 0.13 nm. After LOW immersion, the well-defined surface covered with single atomic step lines and atomically flat terraces were formed as shown in Fig. 1 [6, 7]. After LOW treatment, a Si wafers were thermally oxidized in 3.8% O₂ diluted with argon at 800 and 1000 °C. After the thermal oxidation, the topography of the SiO₂ surfaces was observed using AFM. The Si/SiO₂ interface topography was observed after removal of SiO₂ films. The RMS value within a 200 x 200 nm² area was used as an indicator of the roughness. On (111)-oriented Si wafers, such well-defined step/terrace structures can be easily obtained. The SiO₂ films were removed to observe the Si/SiO₂ interface structure. The Si(111) surface was found to be sufficiently stable against attack by this solution, but the Si(100) surface was not. Therefore, the (111)-oriented wafers were used in this work.

Results and Discussion

Figure 2 shows typical AFM images of the surface and interface of 2.1-, 5.4-, and 12.6 nm-thick SiO₂ grown at 800 °C in 3.8% O₂ diluted with argon. Profiles along the solid lines in each image are also shown at the same height scale of 0.75 nm. It is seen that with increasing film thickness, the roughness at both the surface and interface increases. Figure 3(a) shows the SiO₂ film thickness

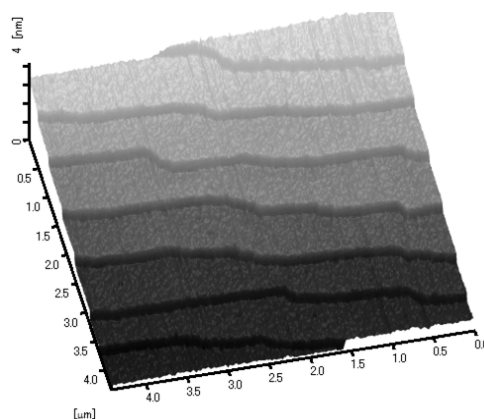


Fig. 1 Well-defined step/terrace structure on a (111)-oriented Si surface after LOW treatment. The typical RMS value within a terrace surface is less than 0.05 nm.

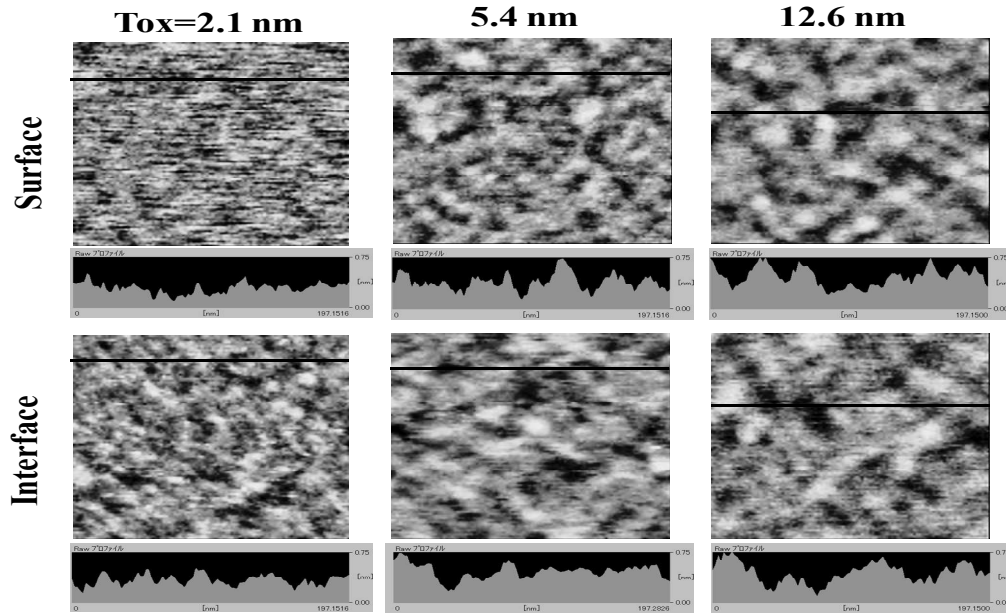


Fig. 2 Typical AFM images of the surface and interface of 2.1-, 5.4-, and 12.6 nm thick SiO_2 films thermally grown at 800 °C in 3.8 % O_2 diluted with argon.

roughness at both the surface and interface increases. Figure 3(a) shows the SiO_2 film thickness dependence of the RMS value at SiO_2 surface. The typical RMS value of an initial atomically flat Si terrace before thermal oxidation was less than 0.05 nm. With increasing film thickness, the RMS value increased almost linearly for thicknesses of less than 5 nm and saturated at a film thickness of approximately 5 nm. The saturation thickness for oxidation at 800 °C is more than that at 600 °C and is less than that at 1000 °C [8]. That is, as the oxidation temperature increased, the roughening rate decreased, and the saturation thickness increases. In Fig. 3(b), the RMS value of the interface is plotted as a function of the film thickness. Although the data is scattered, the RMS value of the interface is also saturated at approximately 5 nm.

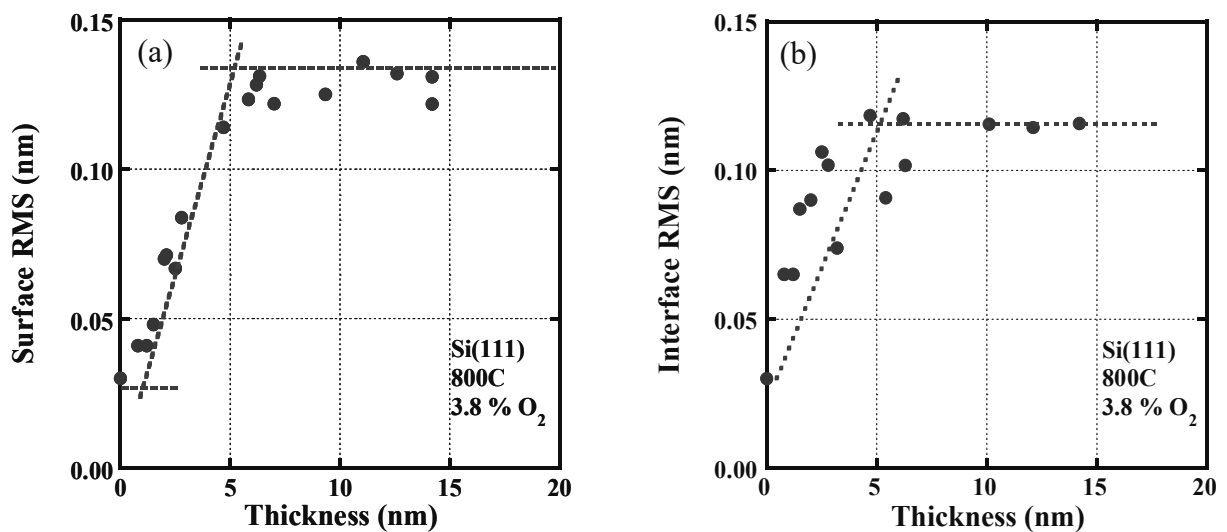


Fig. 3 (a) Surface RMS value and (b) interface RMS value plots as a function of thickness of the SiO_2 films grown at 800 °C in 3.8 % O_2 diluted with argon.

In Fig. 4, the relation between film thickness and oxidation time is plotted in the case for thermal oxidation at 800 °C in 3.8% O₂ ambient. The thickness of SiO₂ thermally grown in the initial oxidation region is approximately 4 nm, which is almost the same as the saturation thickness described above. The initial oxidation thickness at 800 °C is less than that at 1000 °C. It is presumed that this is related to the diffusion length of the SiO emitted from the Si/SiO₂ interface [9,10] to relax the oxidation stress. The steep increase of film thickness in the initial oxidation region was simulated taking into account the re-oxidation of the emitted SiO. The correlation between the roughness saturation thickness and the initial oxidation thickness indicated that the generation and growth of the roughness is closely related to the re-oxidation of the SiO emitted from the Si/SiO₂ interface[11].

In the LOW treatment, an SiO₂ on the Si surface is not attacked [12]. We formed a step/terrace structure within the Si surface surrounded by a thick SiO₂ film using photo-lithography. The SiO₂ pattern was used as a guide to observe the surface morphology at the same position on the SiO₂ film thermally grown on the Si terrace.

Figure 5 shows typical surface AFM images of four thicknesses of SiO₂ thermally grown on an atomically flat silicon terrace. By using the SiO₂ patterns and step line patterns as two guides, we can recognize that ongoing thermal oxidation changes the surface morphology of the SiO₂ in the same region. When the formed SiO₂ is thin, the step line patterns are taken over from the initial silicon surface to the SiO₂ surface. A number of the protuberances are generated just after oxidation starts. Subsequently, the height of the protuberances is increased by oxidation, with almost no change in their locations.

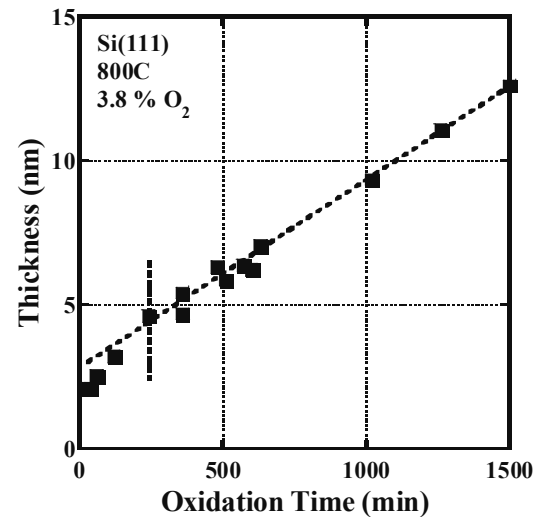


Fig. 4 SiO₂ film thickness as a function of oxidation time at 800 °C in 3.8 % O₂, diluted by argon.

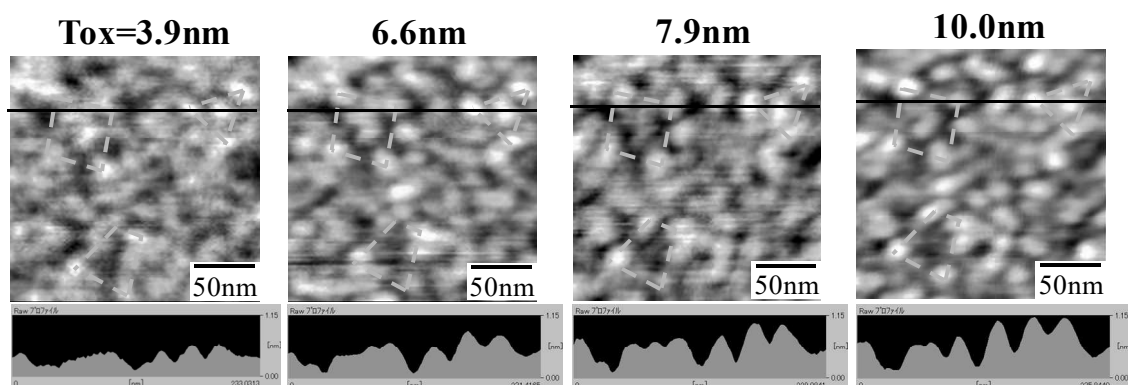


Fig. 5 AFM images of the surfaces of SiO₂ grown at 1000 °C in 3.8 % O₂ ambient. The film thicknesses are 3.9-, 6.6-, 7.9-, and 10.0-nm. The triangle and quadrangles in each figure indicate the same set of the protuberances.

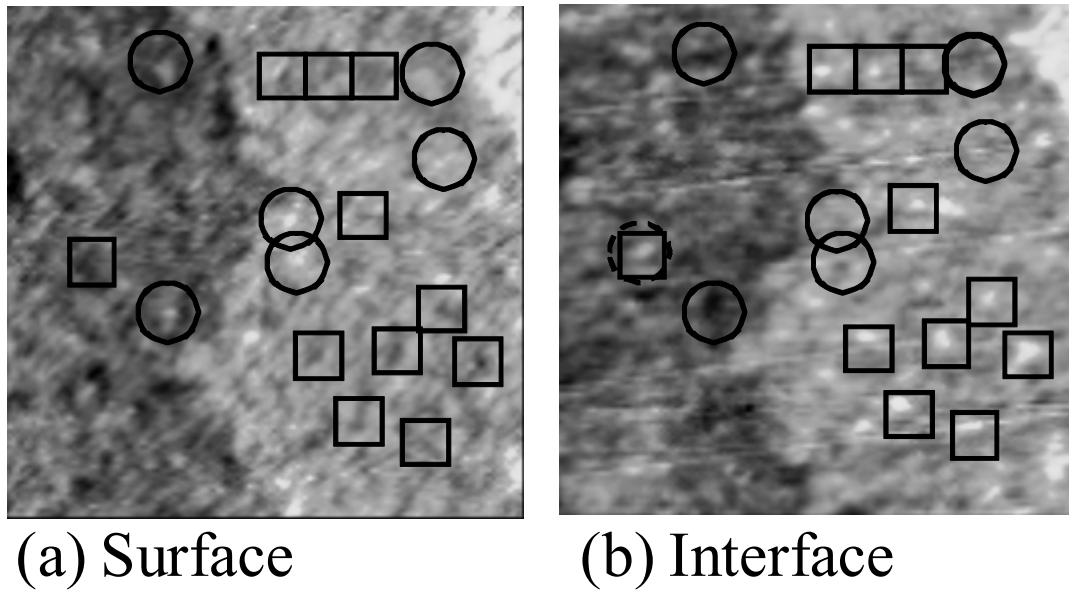


Fig. 6 Typical AFM images of the surface and interface in the same area of 500 nm x 500 nm square area. Circles and squares on the corresponding positions at the surface and interface are drawn.

Using a technique wherein surface topographies of SiO_2 with various film thicknesses can be observed in the same region, the two-dimensional correlation of topography between the SiO_2 surface and Si/SiO_2 interface needs to be investigated to clarify local thinning of thermally grown SiO_2 films. In Figs. 6(a) and 6(b), the typical AFM images of the surface and interface of SiO_2 with a thickness of 5.6 nm are shown. Based on the zigzag step line pattern in this figure, it is confirmed that the images are of the surface and interface of the SiO_2 in the same region. The circles and squares in both images are drawn the corresponding locations. The circles and squares in Fig. 6(a) indicate the positions of protuberances and dimples on the SiO_2 surface image, respectively. It can be noticed that in these positions the protuberances on the SiO_2 surface correspond to the dimples at the Si/SiO_2 interface and vice versa. Since it is confirmed that the immersion in diluted hydrofluoric acid (d-HF) maintains the atomic structure of the Si surface, the dimples appear after removal of the formed SiO_2 by d-HF immersion. At the Si/SiO_2 interface, the dimples indicate the occurrence of local oxidation. The correspondence between the position of the protuberances at the surface and the dimples at the interface indicates is one of origins of the protuberances at the SiO_2

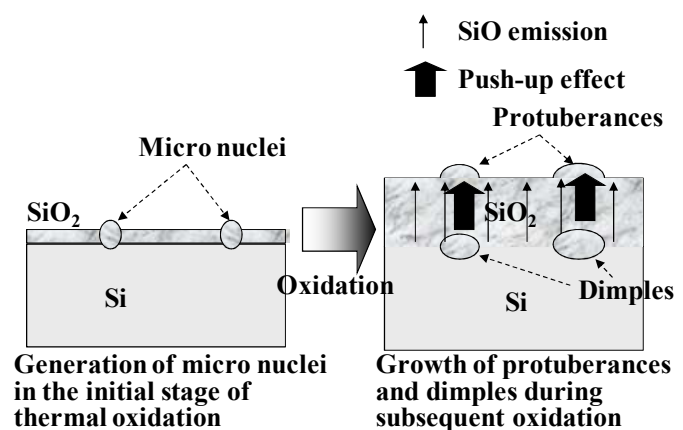


Fig. 7 A model of nonuniform oxidation. Two mechanisms of the SiO emission from the Si/SiO_2 interface and the push-up of the SiO_2 by the local oxidation at the interface contribute to the protuberance growth at the SiO_2 surface.

surface. As shown in Fig.7, micro nuclei which are the origins of the protuberances at the surface and the dimples at the interface are generated immediately after the start of the oxidation. The preexisting thin SiO₂ film is then pushed up by the oxide growing locally growing at the interface. In other words, the correspondence in Fig. 6 indicates that the position of local oxidation by the multilayer oxidation is determined at the initial stage of the oxidation. During subsequent oxidation, the position of the preceding local oxidation is maintained. However, the push-up effect alone cannot explain the saturation of the surface RMS value shown in Fig.3. Re-oxidation of the SiO emitted from the interface should also be taken into account. A comparison of Fig. 3 described above and Figs. 3 and 6 in Ref. 11 suggests that the relative contribution of each mechanism depends on the oxidation condition used.

On the other hand, the correspondence of the dimples at the surface and the protuberances at the interface indicates that the oxidation is locally delayed there. That is, the local thinning of the SiO₂ film occurs. This influences the dielectric breakdown characteristics of the SiO₂ films.

Summary

Precise AFM observations of the surface and interface of SiO₂ thermally grown at 800 °C in 3.8 % O₂ diluted with argon at the atmospheric pressure on atomically flat Si surface have been carried out to determine the SiO₂ thickness uniformity on an atomic scale. Protuberances generated at the initial stage of oxidation were found to increase in height as oxidation proceeds. The surface RMS value of SiO₂ was found to increase linearly up to a thickness of approximately 5 nm, which almost corresponds to the thickness of the initial oxidation layer. This fact indicates that the roughness increase is related to re-oxidation of SiO emitted from the Si/SiO₂ interface during the initial oxidation. In addition, the surface RMS value saturated at an RMS value of approximately 0.13 nm. A correspondence between the position of dimples at the surface and the positions of the protuberances at the interface was also observed. This is the direct evidence of the local thinning of the thermally grown SiO₂ films. These experimental results were clarified for a wide atomically flat silicon terrace after LOW immersion.

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